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# From two-dimensional graphene oxide to three-dimensional honeycomb-like Ni<sub>3</sub>S<sub>2</sub>@graphene oxide composite: insight into structure and electrocatalytic properties

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Three-dimensional (3D) graphene composites have drawn increasing attention in energy storage/conversion applications due to their unique structures and properties. Herein, we synthesized 3D honeycomb-like Ni<sub>3</sub>S<sub>2</sub>@graphene oxide composite (3D honeycomb-like Ni<sub>3</sub>S<sub>2</sub>@GO) by a one-pot hydrothermal method. We found that positive charges of Ni<sup>2+</sup> and negative charges of NO<sub>3</sub><sup>-</sup> in Ni(NO<sub>3</sub>)<sub>2</sub> induced a transformation of graphene oxide with smooth surface into graphene oxide with wrinkled surface (w-GO). The w-GO in the mixing solution of Ni(NO<sub>3</sub>)<sub>2</sub>/thioacetamide/H<sub>2</sub>O

evolved into 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  in solvothermal process. The GO effectively inhibited the aggregation of  $\text{Ni}_3\text{S}_2$  nanoparticles. Photoelectrochemical cells based on 3D  $\text{Ni}_3\text{S}_2@\text{GO}$  synthesized at  $60\text{ mM l}^{-1}$   $\text{Ni}(\text{NO}_3)_2$  exhibited the best energy conversion efficiency. 3D  $\text{Ni}_3\text{S}_2@\text{GO}$  had smaller charge transfer resistance and larger exchange current density than pure  $\text{Ni}_3\text{S}_2$  for iodine reduction reaction. The cyclic stability of 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  was good in the iodine electrolyte. Results are of great interest for fundamental research and practical applications of 3D GO and its composites in solar water-splitting, artificial photoelectrochemical cells, electrocatalysts and Li-S or Na-S batteries.

## 1. Introduction

Graphene is composed of  $\text{SP}^2$  hybrid C atoms. The thickness of two-dimensional (2D) monolayer graphene is  $3.35\text{ \AA}$ . Six C atoms in the same plane bond with the adjacent C atoms in the sigma format, which makes the graphene have good structural rigidity. The orbit of surplus p is perpendicular to the graphene plane. The big  $\pi$  bonds are formed by overlapping of other p orbits. The electrons in big  $\pi$  bonds can move freely, which gives good conductivity to graphene. The unique monolayer graphene has a large theoretical specific surface area of about  $2630\text{ m}^2\text{ g}^{-1}$ . It also has high conductivity, high electron mobility ( $15\,000\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$ ) [1], and thermal conductivity, quantum Holzer effect, quantum tunnelling effect [2], super mechanical properties [3], and so on. However, it lacks semiconductor properties, which limits its application in many fields.

Three-dimensional (3D) graphene composites have drawn increasing attention in energy storage/conversion applications due to their unique structures and properties [4–9]. Freestanding, lightweight 3D graphene networks as ultralight and flexible supercapacitor electrodes were prepared from pressed Ni foam [10]. A 3D graphene-based hierarchically porous carbon has been prepared by a dual template strategy and explored as an electrode for capacitive deionization [11]. A 3D carbon fibre/reduced graphene oxide composite textile was prepared by introducing 2D reduced graphene oxide interfaces into one-dimensional (1D) carbon fibre networks [12].

The composites of graphene and transition metal complexes (TMC) possess electrical conductivity, thermal conductivity, structure stability and excellent catalytic activity. The structures of TMC/graphene composite material have four categories: 0D/2D (zero-dimensional TMC/two-dimensional graphene), 1D/2D (one-dimensional TMC/two-dimensional graphene), 2D/2D (two-dimensional TMC/two-dimensional graphene) and 3D (TMC/three-dimensional graphene).

3D NiS/G composite and 3D CoS/G composite were prepared by three steps: growing graphene by chemical vapour deposition, coating precursors and further annealing [13]. The dye-sensitized solar cells (DSCs) with 3D NiS/G composite and 3D CoS/G composite counter electrodes showed good electrocatalytic activity and photovoltaic conversion efficiencies of 5.04% and 5.25%, respectively. We previously reported that unique ZnS nanobuns decorated with 2D reduced graphene oxide were synthesized using one-pot solvothermal method. Graphene in ZnS@GO remained in 2D structures (flake-like shape with few wrinkles). Herein, we synthesized 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{graphene oxide}$  composite (3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$ ) by a one-pot hydrothermal method. We study the structural characteristics of 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  evolved from 2D graphene oxide. The electrocatalytic characteristics of 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  for iodine reduction reaction are investigated by electrochemical impedance (EIS) and Tafel polarization. The effects of 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  structure on photovoltaic parameters are also investigated.

## 2. Material and methods

### 2.1. Preparation of pure $\text{Ni}_3\text{S}_2$ block

Typically,  $0.0017\text{ mol Ni}(\text{NO}_3)_2\cdot 6\text{H}_2\text{O}$  was dissolved in 30 ml deionized water by vigorous agitation. Thioacetamide (TAA) ( $0.023\text{ mol}$ ) was dissolved in 30 ml deionized water by vigorous agitation. The above two solutions were mixed. The mixture was stirred for 0.5 h at room temperature and then transferred into a Teflon-lined autoclave. After being heated at  $200^\circ\text{C}$  for 24 h, the product was cooled to room temperature naturally. The product was washed three times with water and ethanol.

## 2.2. Preparation of three-dimensional honeycomb-like Ni<sub>3</sub>S<sub>2</sub>@graphene oxide

Specific synthesis method of GO could be seen in our previous literature [14–16]. GO (1.125 g 1 wt%) was dispersed in 10 ml deionized water by ultrasound. 0.0017 mol Ni(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O was dissolved in 30 ml deionized water by vigorous agitation. TAA (0.023 mol) was dissolved in 20 ml deionized water by vigorous agitation. The above three solutions were mixed. The mixture was stirred for 0.5 h at room temperature and then transferred into a Teflon-lined autoclave. After being heated at 200°C for 24 h, the product was cooled to room temperature naturally. The product was washed three times with water and ethanol.

## 2.3. Photoanode preparation and cell fabrication

A 12 μm thick layer was deposited on fluorine-doped tin oxide glass by printing 20 nm-sized TiO<sub>2</sub> particles (P25, Degussa, Germany) [15,17]. The obtained film was sintered at 500°C. After cooling to 90°C, the TiO<sub>2</sub> films were immersed in a solution of N719 dye (5 × 10<sup>-4</sup> M) in acetonitrile/*tert*-butyl alcohol (1 : 1 volume ratio) for 20 h. The triiodide/iodide electrolyte for cell testing includes LiI (0.03 M), 1-butyl-3-methylimidazolium iodide (0.6 M), I<sub>2</sub> (0.03 M), 4-*tert*-butylpyridine (0.5 M), guanidinium thiocyanate in acetonitrile (0.1 M). DSCs were assembled by a TiO<sub>2</sub> photoanode with the corresponding counter electrode sandwiching the redox couple in the electrolyte. Symmetrical cells with an effective area of 0.64 cm<sup>2</sup> were analysed by a Tafel-polarization test and by EIS experiments.

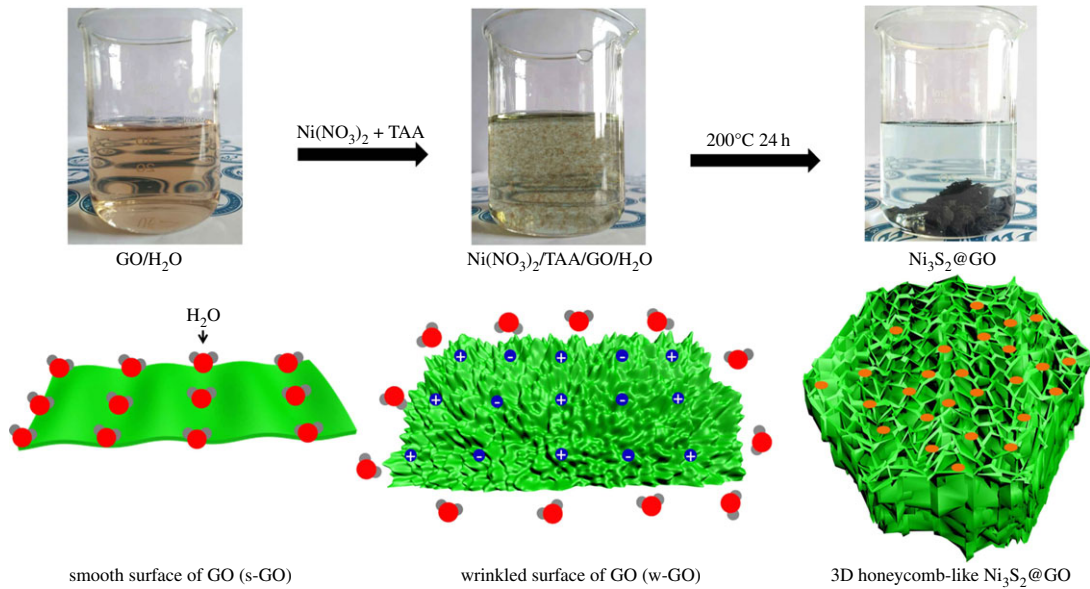
## 2.4. Characterization

To analyse as-synthesized composite electrocatalyst, X-ray diffraction (XRD) patterns were acquired using a PANalytical X'Pert diffractometer (Cu Kα radiation at λ = 1.5406 Å) sampling at 5° min<sup>-1</sup>, 36 kV and 20 mA. As-prepared micro- or nanostructures were characterized and analysed by scanning electron microscopy (SEM; Nova Nano SEM 450). The photocurrent–voltage performance of DSCs with 0.16 cm<sup>2</sup> photoanode film was measured without metal mask by a Keithley digital source meter (Keithley 2400, USA) equipped with a solar simulator (IV5, PV Measurements, Inc., USA). EIS and Tafel experiments were done with symmetrical electrodes in the dark using an electrochemical workstation (CHI760 Chenhua, China). Cyclic voltammetry (CV) was performed in a three-electrode configuration. The triiodide/iodide electrolyte for CV testing includes LiI (2 mM), LiClO<sub>4</sub> (20 mM) and I<sub>2</sub> (0.2 mM).

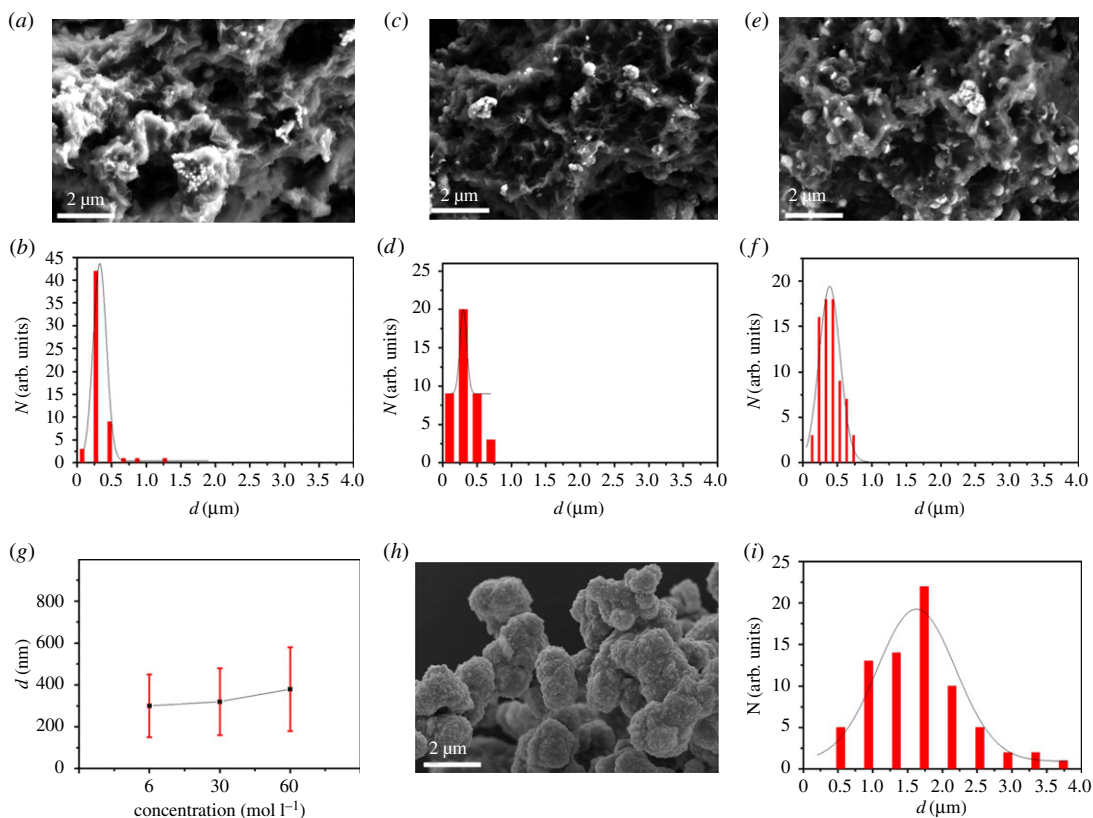
# 3. Results and discussion

The graphene oxide prepared by the Hummers method usually contains a large number of functional groups. The functional groups (hydroxyl and carboxyl groups) on the graphene surface can make graphene more compact and stable when graphene is combined with other materials. As shown in figure 1, graphene oxide prepared by us can be well dispersed in H<sub>2</sub>O to form a homogeneous dispersion. The graphene oxide in H<sub>2</sub>O (GO/H<sub>2</sub>O) can be stable for several months or longer without delamination. The reason for this is the strong hydrogen bonding between hydroxyl and carboxyl groups on the surface of graphene oxide and H<sub>2</sub>O. Therefore, the surface of the graphene oxide in H<sub>2</sub>O is relatively smooth, rather than wrinkled. The graphene oxide with a smooth surface is s-GO. When TAA was added into GO/H<sub>2</sub>O, no obvious change was observed in the homogeneous dispersion. However, when Ni(NO<sub>3</sub>)<sub>2</sub> solution or mixed solution of Ni(NO<sub>3</sub>)<sub>2</sub> and TAA were added into GO/H<sub>2</sub>O, homogeneous dispersion shows lots of flocculent graphene oxide immediately, as shown in figure 1. Because lots of positive charges of Ni<sup>2+</sup> and negative charges of NO<sub>3</sub><sup>-</sup> in Ni(NO<sub>3</sub>)<sub>2</sub> destroy the hydrogen bonding between graphene oxide and H<sub>2</sub>O, the surface of flocculent graphene oxide should be wrinkled rather than smooth, as shown in figure 1. The graphene oxides with wrinkled surface were named w-GO. SEM of w-GO is shown in electronic supplementary material, figure S1. w-GO has two advantages: on the one hand, it can keep the 2D structure of graphene; on the other hand, the wrinkles on the surface can increase the physical and chemical properties in graphene devices. The w-GO suspension is transferred into a Teflon-lined autoclave. After crystallizing and cross-linking of w-GO suspension at 200°C for 24 h, 3D honeycomb-like structures of Ni<sub>3</sub>S<sub>2</sub>@GO were formed, as shown in figure 1.

We investigated the effect of different concentrations of Ni(NO<sub>3</sub>)<sub>2</sub> on the morphology of Ni<sub>3</sub>S<sub>2</sub>@GO. As can be seen from figure 2, the microstructures of all the synthesized Ni<sub>3</sub>S<sub>2</sub>@GO samples obtained using 6, 30 and 60 mM Ni(NO<sub>3</sub>)<sub>2</sub> have 3D honeycomb-like structures. This kind of 3D honeycomb-like structure possessed larger specific surface area and more surface catalytic activity sites. In addition,

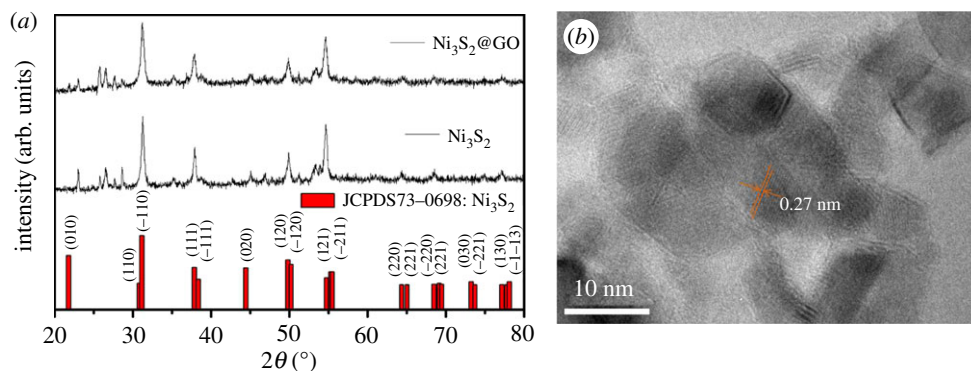


**Figure 1.** Picture and structural evolution of 3D honeycomb-like  $\text{Ni}_3\text{S}_2@GO$  from graphene oxide with smooth surface and graphene oxide with wrinkled surface.



**Figure 2.** (a,b) SEM image and size distribution for 3D honeycomb-like  $\text{Ni}_3\text{S}_2@GO$  synthesized under  $6 \text{ mM l}^{-1} \text{ Ni}(\text{NO}_3)_2$ ; (c,d) SEM image and size distribution for 3D honeycomb-like  $\text{Ni}_3\text{S}_2@GO$  synthesized under  $30 \text{ mM l}^{-1} \text{ Ni}(\text{NO}_3)_2$ ; (e,f) SEM image and size distribution for 3D honeycomb-like  $\text{Ni}_3\text{S}_2@GO$  synthesized under  $60 \text{ mM l}^{-1} \text{ Ni}(\text{NO}_3)_2$ ; (g) the relationship between the size of  $\text{Ni}_3\text{S}_2$  on graphene surface and the concentration of  $\text{Ni}(\text{NO}_3)_2$ ; (h,i) SEM image and size distribution for pure  $\text{Ni}_3\text{S}_2$  synthesized under  $30 \text{ mM l}^{-1} \text{ Ni}(\text{NO}_3)_2$ .

the coupling between the walls of w-GO and  $\text{Ni}_3\text{S}_2$  will play the role of synergistic catalysis. There are few  $\text{Ni}_3\text{S}_2$  nanoparticles on the 3D honeycomb-like  $\text{Ni}_3\text{S}_2@GO$  under  $6 \text{ mM l}^{-1} \text{ Ni}(\text{NO}_3)_2$ . The size of  $\text{Ni}_3\text{S}_2$  nanoparticles on the 3D honeycomb-like  $\text{Ni}_3\text{S}_2@GO$  is approximately 300 nm (figure 2a,b). When the concentration of  $\text{Ni}(\text{NO}_3)_2$  is  $30 \text{ mM l}^{-1}$ , the numbers of  $\text{Ni}_3\text{S}_2$  nanoparticles on the 3D



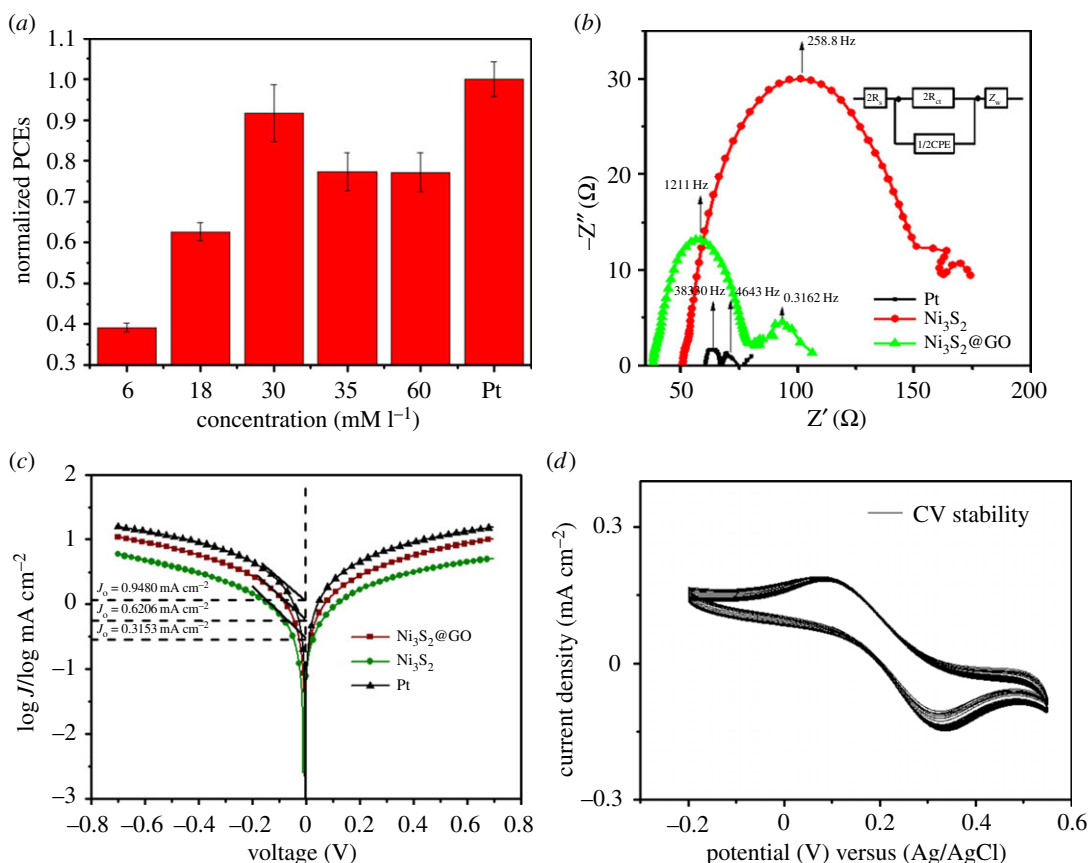
**Figure 3.** (a) XRD patterns of as-prepared 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  and pure  $\text{Ni}_3\text{S}_2$  synthesized under  $30 \text{ mM l}^{-1} \text{ Ni}(\text{NO}_3)_2$ ; (b) high-resolution transmission electron micrograph of as-prepared 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  under  $30 \text{ mM l}^{-1} \text{ Ni}(\text{NO}_3)_2$ .

honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  are increased. The size of  $\text{Ni}_3\text{S}_2$  nanoparticles on the 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  is approximately 320 nm, as shown in figure 2c,d. The numbers of the  $\text{Ni}_3\text{S}_2$  nanoparticles on the 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  under  $60 \text{ mM l}^{-1}$  are more than those under 6 and  $30 \text{ mM l}^{-1}$   $\text{Ni}(\text{NO}_3)_2$ . The size of  $\text{Ni}_3\text{S}_2$  nanoparticles on the 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  is approximately 380 nm (figure 2e,f). As shown in figure 2g, the size of graphene surface  $\text{Ni}_3\text{S}_2$  is gradually increased with increasing concentration of  $\text{Ni}(\text{NO}_3)_2$ . The SEM results of pure  $\text{Ni}_3\text{S}_2$  material synthesized under  $30 \text{ mM l}^{-1}$   $\text{Ni}(\text{NO}_3)_2$  are shown in figure 2h. It can be seen that the pure  $\text{Ni}_3\text{S}_2$  block was larger than  $\text{Ni}_3\text{S}_2$  in  $\text{Ni}_3\text{S}_2@\text{GO}$ . The size of  $\text{Ni}_3\text{S}_2$  block is approximately  $1.7 \mu\text{m}$ , as shown in figure 2i. The results suggested that the presence of graphene significantly inhibits aggregation of  $\text{Ni}_3\text{S}_2$  nanoparticles, which is consistent with our previous findings [11].

The XRD pattern of the 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  powder is shown in figure 3a. In order to classify the diffraction peaks, we synthesized pure  $\text{Ni}_3\text{S}_2$  as the contrast. The diffraction pattern of 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  is consistent with that of pure  $\text{Ni}_3\text{S}_2$ . All of the peaks are in accordance with the  $\text{Ni}_3\text{S}_2$  standard card (PDF#73-0698). The strong diffraction peaks at  $31.26^\circ$  and  $54.7^\circ$  corresponded to  $(-110)$  and  $(-211)$  crystal planes. The other diffraction peaks at  $23.04^\circ$ ,  $37.94^\circ$  and  $49.88^\circ$  corresponded to  $(010)$ ,  $(111)$ ,  $(120)$  crystal planes. From further structural analysis by high-resolution transmission electron microscopy, the lattice spacing of  $\text{Ni}_3\text{S}_2$  on  $\text{Ni}_3\text{S}_2@\text{GO}$  is 0.27 nm, which is close to  $[-110]$  spacing of  $\text{Ni}_3\text{S}_2$  (PDF#73-0698).

3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  synthesized by different concentrations of  $\text{Ni}(\text{NO}_3)_2$  was prepared in thin film by the spraying method and used as a counter electrode in DSCs. Meanwhile, pyrolytic platinum was prepared to be used as a reference. The normalized power conversion efficiency is shown in figure 4a. The results indicated that the highest energy conversion efficiency is based on 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  synthesized by  $30 \text{ mM l}^{-1}$   $\text{Ni}(\text{NO}_3)_2$ . The effect of 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  structure synthesized by different  $\text{Ni}(\text{NO}_3)_2$  concentration on open circuit voltage ( $V_{oc}$ ), short circuit current density ( $J_{sc}$ ) and fill factor is shown in electronic supplementary material, figures S2, S3 and S4. The best energy conversion efficiency based on 3D  $\text{Ni}_3\text{S}_2@\text{GO}$  synthesized at  $60 \text{ mM l}^{-1}$   $\text{Ni}(\text{NO}_3)_2$  is shown in electronic supplementary material, figure S5. To investigate the reason for the good performance of 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  materials in DSCs, EIS and Tafel polarization were carried out. EIS is an electrochemical method widely used for the characterization of counter electrode. Figure 4b is a typical Nyquist diagram. Each Nyquist diagram usually consists of two semicircles. The resistance of left semicircle starting on X-axis represented the series resistance ( $R_s$ ). The value of left semicircle diameter represented charge transfer resistance ( $R_{ct}$ ) between the electrode and electrolyte.  $R_{ct}$  occurs in the high-frequency region and is closely related to the electrocatalytic properties. According to the symmetrical cell equivalent circuit diagram in figure 4b, the EIS parameters obtained by Z-View software are listed in table 1.

The values of  $R_s$  for 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  and pure  $\text{Ni}_3\text{S}_2$  in  $\text{I}^-/\text{I}_3^-$  electrolyte system are  $38.19 \Omega$  and  $50.72 \Omega$ , respectively. The main reason for the smaller  $R_s$  of 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  is the high conductivity of graphene in composites. The value of  $R_{ct}$  for 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  is  $40.29 \Omega$ , which is much smaller than that of pure  $\text{Ni}_3\text{S}_2$  ( $106.4 \Omega$ ). The results indicated that 3D honeycomb-like  $\text{Ni}_3\text{S}_2@\text{GO}$  exhibited better electrocatalytic activity than pure  $\text{Ni}_3\text{S}_2$  for the reduction reaction of iodine. The frequency at highest point of left semicircle for  $\text{Ni}_3\text{S}_2$  is



**Figure 4.** (a) The normalized power conversion efficiency (PCE) based on 3D honeycomb-like  $\text{Ni}_3\text{S}_2@/\text{GO}$  synthesized by different concentration of  $\text{Ni}(\text{NO}_3)_2$ ; (b) EIS of symmetrical cells fabricated with two identical 3D honeycomb-like  $\text{Ni}_3\text{S}_2@/\text{GO}$  or pure  $\text{Ni}_3\text{S}_2$  (synthesized by 30  $\text{mM l}^{-1}$   $\text{Ni}(\text{NO}_3)_2$ ) under the bias voltage with open voltage corresponding to photovoltaic devices; (c) Tafel polarization curves of symmetrical cells fabricated with two identical 3D honeycomb-like  $\text{Ni}_3\text{S}_2@/\text{GO}$  or pure  $\text{Ni}_3\text{S}_2$  (synthesized by 30  $\text{mM l}^{-1}$   $\text{Ni}(\text{NO}_3)_2$ ) under the bias voltage with open voltage corresponding to photovoltaic devices; (d) the cyclic stability of 3D honeycomb-like  $\text{Ni}_3\text{S}_2@/\text{GO}$  (synthesized by 30  $\text{mM l}^{-1}$   $\text{Ni}(\text{NO}_3)_2$ ) in the iodine electrolyte.

**Table 1.** Series resistance ( $R_s$ ), charge transfer resistance ( $R_{ct}$ ) and exchange current density based on the symmetrical cells of 3D honeycomb-like  $\text{Ni}_3\text{S}_2@/\text{GO}$ ,  $\text{Ni}_3\text{S}_2$  (synthesized by 30  $\text{mM l}^{-1}$   $\text{Ni}(\text{NO}_3)_2$ ) and Pt in the iodine electrolyte.

| electrolytes              | CEs                                | $R_s/\Omega$ | $R_{ct}/\Omega$ | $J_0/\text{mA cm}^{-2}$ |
|---------------------------|------------------------------------|--------------|-----------------|-------------------------|
| $\text{I}^-/\text{I}_3^-$ | $\text{Ni}_3\text{S}_2@/\text{GO}$ | 38.19        | 40.29           | 0.6206                  |
|                           | $\text{Ni}_3\text{S}_2$            | 50.72        | 106.4           | 0.3153                  |
|                           | Pt                                 | 60.86        | 6.866           | 0.9480                  |

258.8 Hz, which corresponds to a time constant of 0.003864 s. The frequency at highest point of left semicircle for 3D honeycomb-like  $\text{Ni}_3\text{S}_2@/\text{GO}$  is 1211 Hz, which corresponds to a time constant of 0.0008921 s. The time constants also indicated that the catalytic performance of 3D honeycomb-like  $\text{Ni}_3\text{S}_2@/\text{GO}$  was better than that of pure  $\text{Ni}_3\text{S}_2$ .  $R_{ct}$  and the exchange current density ( $J_0$ ) in Tafel was inversely proportional, according to  $R_{ct} = RT/nFJ_0$ . The values of  $J_0$  for 3D honeycomb-like  $\text{Ni}_3\text{S}_2@/\text{GO}$  and pure  $\text{Ni}_3\text{S}_2$  are 0.6317  $\text{mA cm}^{-2}$  and 0.36625  $\text{mA cm}^{-2}$ , respectively. The result indicated that the iodine reduction reaction has faster electron exchange on the surface of 3D honeycomb-like  $\text{Ni}_3\text{S}_2@/\text{GO}$ , which is consistent with the EIS results. The cyclic stability of 3D honeycomb-like  $\text{Ni}_3\text{S}_2@/\text{GO}$  in the iodine electrolyte is shown in figure 4d. The current density and the potential position had no significant change, which indicated the good stability of 3D honeycomb-like  $\text{Ni}_3\text{S}_2@/\text{GO}$  in the iodine electrolyte.

## 4. Conclusion

In sum, 3D honeycomb-like Ni<sub>3</sub>S<sub>2</sub>@GO was synthesized by a one-pot hydrothermal method. The positive charges of Ni<sup>2+</sup> and negative charges of NO<sub>3</sub><sup>-</sup> in Ni(NO<sub>3</sub>)<sub>2</sub> induced a transformation of s-GO into w-GO. The GO can effectively inhibit the aggregation of Ni<sub>3</sub>S<sub>2</sub> nanoparticles. 3D honeycomb-like Ni<sub>3</sub>S<sub>2</sub>@GO exhibited good electrocatalytic activity and photoelectrochemical performance. These findings are of great interest for fundamental research and practical applications of 3D graphene oxides and their composites.

Data accessibility. Additional data are in the electronic supplementary material.

Authors' contributions. J.Y. and H.Z. designed the study, interpreted the results and wrote the manuscript. X.W., Y.L., W.X., K.Z., S.S., J.W., F.D., J.G., C.W. and C.C. prepared all samples for analysis. N.S., B.C., Y.Z., H.H., X.Z., J.Z. and S.W. collected and analysed the data. All authors gave their final approval for publication.

Competing interests. We declare we have no competing interests.

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