

Research Article

Nanostructural Organization of Naturally Occurring Composites—Part I: Silica-Collagen-Based Biocomposites

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Glass sponges, as examples of natural biocomposites, inspire investigations aiming at both a better understanding of biomineralization mechanisms and novel developments in the synthesis of nanostructured biomimetic materials. Different representatives of marine glass sponges of the class Hexactinellida (Porifera) are remarkable because of their highly flexible basal anchoring spicules. Therefore, investigations of the biochemical compositions and the micro- and nanostructure of the spicules as examples of naturally structured biomaterials are of fundamental scientific relevance. Here we present a detailed study of the structural and biochemical properties of the basal spicules of the marine glass sponge *Monorhaphis chuni*. The results show unambiguously that in this glass sponge a fibrillar protein of collagenous nature is the template for the silica mineralization in all silica-containing structural layers of the spicule. The structural similarity and homology of collagens derived from *M. chuni* spicules to other sponge and vertebrate collagens have been confirmed by us using FTIR, amino acid analysis and mass spectrometric sequencing techniques. We suggest that nanomorphology of silica formed on proteinous structures could be determined as an example of biodirected epitaxial nanodistribution of amorphous silica phase on oriented fibrillar collagen templates. Finally, the present work includes a discussion relating to silica-collagen-based hybrid materials for practical applications as biomaterials.

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1. INTRODUCTION

Glass sponges (Hexactinellida: Porifera) provide an abundant source of unusual skeleton structures, which could be defined as natural silica-based nanostructured composite materials. They are intriguing research objects because of the hierarchical organization of their spicules from the nanoscale to the macroscale [1–3]. First observations reported by Lévi et al. [4] on silica-based spicules of a *Monorhaphis* sponge generated great interest because of their combination of properties, namely, toughness combined with stiffness, and resilience. This sponge species synthesizes the largest biosilica structures on earth [5]. Pencil-sized rod spicules, a meter or more in length, could be bent into a circle without breaking. When the load was released, the spicule recovered

its original shape. When the bending of the spicule rod was compared with that of a synthetically derived pure silica rod, the toughness of the spicule was found to be nearly an order of magnitude higher [2]. Recently, the micromechanical properties of biological silica in the giant anchor spicule of *Monorhaphis chuni* were reported on [6]. Nanoidentation showed a considerably reduced stiffness of the spicule compared to technical quartz glass with different degrees of hydration. Moreover, stiffness and hardness were shown to oscillate as a result of the laminate structure of the spicules. Raman spectroscopic imaging showed that the organic layers are protein-rich and that there is an OH-enrichment in silica near the central axial filament of the spicule. Small-angle X-ray scattering revealed the presence of nanospheres with a diameter of only 2.8 nm as the basic unit of silica.

It was suggested that biogenic silica formed by glass sponges possesses reduced stiffness but substantially higher toughness than technical glass due to its architecture, determined by structure at the nanometer and the micrometer level [6]. Unfortunately, the nature and the origin of the protein matrix were not investigated in this study.

There is no doubt that glass sponge anchoring spicules are remarkable objects because of their size, durability, high flexibility, and their exceptional fibre-optic properties, which all together render them of interest as novel biomimetic materials [7]. Of course, the materials science aspects of glass sponges can be studied by model systems, and utilized for biomimetic engineering. However, we cannot mimic nature with a view to designing novel biomaterials without knowledge of the nature and origin of the organic nanostructured matrices of corresponding natural biocomposites which are present in these sponges. Therefore, the biggest shortcoming common to all publications relating to mechanical [2], structural [3], and optical [8] properties of glassy sponge skeletal formations is a lack of real information regarding the chemical nature of corresponding organic matrices.

The finding of collagen within basal spicules of the glass sponge *Hyalonema sieboldi* [9–11], as well as the occurrence of chitin within the framework skeleton of the glass sponges *Farrea occa* [12], and spicules of *Euplectella aspergillum* [7] as revealed by gentle desilicification in alkali, stimulated further attempts to search for materials of organic nature in skeletal structures of these unique deep-sea organisms. Consequently, the objective of the current study was to test our hypothesis that collagen is also an essential component of the giant anchoring silica spicules of *Monorhaphis chuni*, and if so, to unravel its involvement in the mechanical behavior of these formations, which was well investigated recently [6].

In the present work, we provide a detailed study confirming our hypothesis that the nanofibrillar organic matrix of collagenous nature within the giant spicules of *M. chuni* is responsible for their extraordinary mechanical properties. We performed structural, spectroscopic, and biochemical analyses of these glassy composites. Finally, this work includes a discussion relating to practical applications of silica-collagen composites artificially derived in vitro as biomaterials for use in biomedicine, engineering, and materials science.

2. EXPERIMENTAL

2.1. Chemical etching of spicules and extraction of collagen

Monorhaphis chuni was collected by the R.V. “Vitiaz-2 (4),” voyage 17, St. 2601, 12° 31.5′–25.04′ S 48° 05.5′–08.0′ E, depth 700 m. Dried *Monorhaphis* basal spicules (length 120 cm, diameter 1.5–4.5 mm, Figure 1) were washed three times in distilled water, cut into 2–5 cm long pieces and placed in a solution containing purified *Clostridium histolyticum* collagenase (Sigma) to digest any possible collagen contamination of exogenous nature. After incubation for 24 hours at 15°C [13], the pieces of spicules were washed again three times in distilled water, dried and placed in 10 ml plastic vessels containing 5 ml of 2.5 M NaOH solution. The

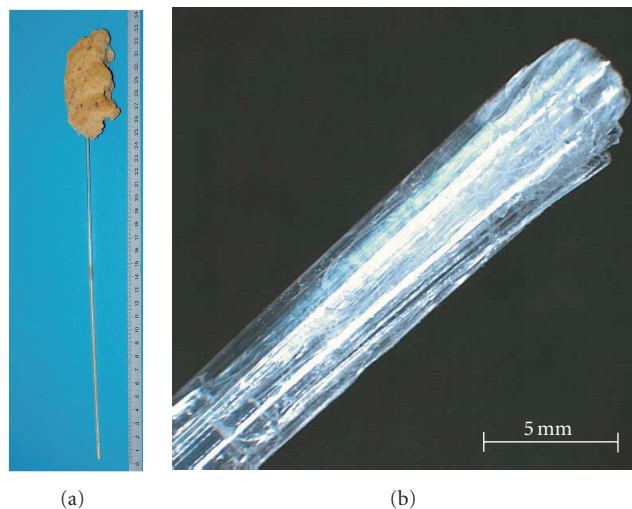


FIGURE 1: (a) Marine glass sponge *Monorhaphis chuni*, a member of the hexactinellids, (b) the sponge consists of a giant basal spicule which anchors *Monorhaphis* to the sandy substratum.

vessel was covered, placed under thermostatic conditions at 37°C and shaken slowly for 14 days. The effectiveness of the slow alkali etching was monitored using scanning electron microscopy (SEM) at different locations along the spicules' length and within the cross-sectional area.

2.2. Biochemical analysis of collagen

Alkali extracts of *Monorhaphis* spicules containing fibrillar protein were dialyzed against deionized water on Roth (Germany) membranes with a cut-off of 14 kDa. Dialysis was performed for 48 hours at 4°C. The dialyzed material was dried under vacuum conditions in a CHRIST lyophilizer (Germany). The approximate molecular weights of proteins in the lyophilizate were determined by gel electrophoresis in the presence of sodium dodecyl sulphate in 10% and 12% gel plates. The kit of molecular weight markers (Silver stain SDS molecular standard mixtures) from Sigma, USA, was used. Lyophilizates were dissolved in sample buffer (1 M Tris-HCl, pH 6.8, 2.5% SDS, 10% glycerine, 0.0125% bromphenol blue) incubated at 95°C for 5 minutes and then applied to 10% or 12% of SDS-polyacrylamide gels. After electrophoresis at 75 V for 1.5 hours, 10% gels were stained with GelCode SilverSNAP Stain Kit II (Pierce, USA), and 12% gels were stained with coomassie brilliant blue R250 to allow proteins to be visualized. To elucidate the nature of proteins isolated from glass sponge spicules, corresponding electrophoretic gels stained with Coomassie were used for the determination of the amino acid sequence by the mass spectrometric sequencing technique (MALDI, Finnigan LTQ) as described earlier [14].

2.3. Structural analysis of spicule layers

Structural analysis of the glass sponge basal spicules and corresponding extracted proteinaceous components was performed using scanning electron microscopy (SEM)

(ESEM XL 30, Philips) and transmission electron microscopy (TEM) (Zeiss EM 912). Additional transmission electron microscopy experiments were carried out at the Special Triebenberg Laboratory for electron holography and high-resolution microscopy of the Technical University Dresden. A field-emission microscope of the FEI company (Endhoven, NL) CM200 FEG/ST-Lorentz was used equipped with a 1×1 k CCD camera (multiscan, Gatan, USA). The analysis of the TEM images was realized by means of the Digital Micrograph software (Gatan, USA). Infrared spectra were recorded with a Perkin Elmer FTIR Spectrometer Spectrum 2000, equipped with an AutoImage Microscope using the fourier transform infrared reflection absorption spectroscopy (FT-IRRAS) technique. In the case of the FTIR-analyses, calf skin collagen (Fluka) and *Chondrosia reniformis* sponge collagen (Klinipharma GmbH, Germany) were investigated as reference samples.

2.4. Silicification of collagen in vitro

Tetramethoxysilan (TMOS 98%, ABCR GmbH, Germany) was chosen as a silica precursor and was hydrolysed for 24 h at 4°C by adding water as well as HCl as a catalyst. This procedure results in the soluble form of silica—orthosilicic acid—whose further polycondensation reactions can be divided into monomer polymerisation, nuclei growth, and aggregation of particles. Hybridization—the combination of silica and collagen—was performed by intensive mixing of prehydrolysed TMOS and the homogeneous collagen suspensions under ambient conditions as described in [15].

2.5. Biocompatibility of the silica-collagen hybrid materials

was evaluated by cultivating human mesenchymal stem cells on the material followed by induced differentiation into osteoblast-like cells [16].

3. RESULTS AND DISCUSSION

It was generally accepted that the skeletons of Hexactinellida are composed of amorphous hydrated silica deposited around a proteinaceous axial filament [17, 18]. The nanolocalization of the proteinaceous component of the glass sponge spicules was not investigated in detail because of lack of a demineralization method which preserved the organic matrix during desilicification. Up to now, the common technique for the desilicification of sponge spicules was based on hydrogen fluoride solutions [5], however this kind of demineralization is rather aggressive chemical procedure which could drastically change the structure of proteins [19, 20]. To overcome this obstacle, Ehrlich et al. [9–11] developed novel, slow etching methods, which use solutions of 2.5 M NaOH at 37°C and take 14 days. Using these methods, it was shown for the first time that the same class of proteins—collagen—involved in cartilage and bone formation also forms the matrix and deposition site of amorphous silica in *H. sieboldi* glass sponge spicules [9, 21]. It was suggested that the *H. sieboldi* basal spicule is an example of a biocomposite con-

taining a silicified collagen matrix and that the high collagen content is the origin of the high mechanical flexibility of the spicules.

SEM investigations of the alkali-etched *Monorhaphis chuni* spicules (Figure 2(a)) confirmed the multilayered silica structure, well-known since the first microscopically investigation of hexactinellid sponges by Schultze in 1860 [22], and present in all representatives of lyssacine Hexactinellida [18]. We focused on the investigation of fibrillar components observable at the sites of interstitial layer fractures within partially desilicified spicules. SEM investigations parallel to the slow etching procedures reveal that a fibrillar organic matrix is the template for silica mineralization. Typical fibrillar formations were observed within the tubular silica structures in all layers starting from the inner axial channel containing axial filament (Figure 3(a)) up to the outermost surface layer of the spicules as shown in Figures 2(b) and 2(c). The fibrils in each cylinder form individual concentric 2D networks with the curvature of the corresponding silicate layers. These layers of about 1 μ m in thickness are connected among each other by protein fibres (Figure 2(a)), which possess a characteristic nanofibrillar organization (Figures 2(b) and 2(d)). Partially desilicified nanofibrillar organic matrix observed on the surface of silica-based inner layers of the demineralized spicule provides strong evidence that silica nanoparticles of diameter about 35 nm are localized on the surface of corresponding nanofibrils (Figures 2(c), 2(e), and Figure 3(b)). This kind of silica nanodistribution is very similar to the silica distribution on the surface of collagen fibrils in the form of nanopearl necklets, firstly observed by us in the glass sponge *H. sieboldi* [21]. We suggest that the nanomorphology of silica on proteinous structures described here could be determined as an example of biodirected epitaxial nanodistribution [23] of the amorphous silica phase on oriented organic fibrillar templates.

The nonsilicified microfibrils of the *M. chuni* axial filament with a diameter of approximately 20–30 nm are organized in bundles with a thickness of 1–2 μ m oriented along the axis of the spicule. They can be easily identified by SEM (Figure 3(a)). The morphology of these microfibrils observed by TEM (Figures 4(a) and 4(b)) is very similar to nonstriated collagen fibrils isolated previously from *H. sieboldi* [9–11, 21] and examined using electron microscopy.

Except for collagen, there are some other possible candidates (e.g., silicateins of axial filaments such as in Demospongiae [24, 25] or as recently reported by Müller et al. [5, 26] in *M. chuni*) which would explain the nature and origin of these fibrillar formations. Therefore, a thorough biochemical analysis of isolated fibrils was performed.

The results of the amino acid analysis of protein extracts isolated from demineralized spicules showed an amino acid content typical for collagens isolated from several sources listed in Figure 4 and also reported previously [21]. The same extracts were investigated using PAG-electrophoresis. Corresponding electrophoretic gels stained with Coomassie were used for the determination of the amino acid sequence by a mass spectrometric sequencing technique as described above. We excised two main bands and digested protein material in-gel with trypsin to obtain tryptic peptide mixtures

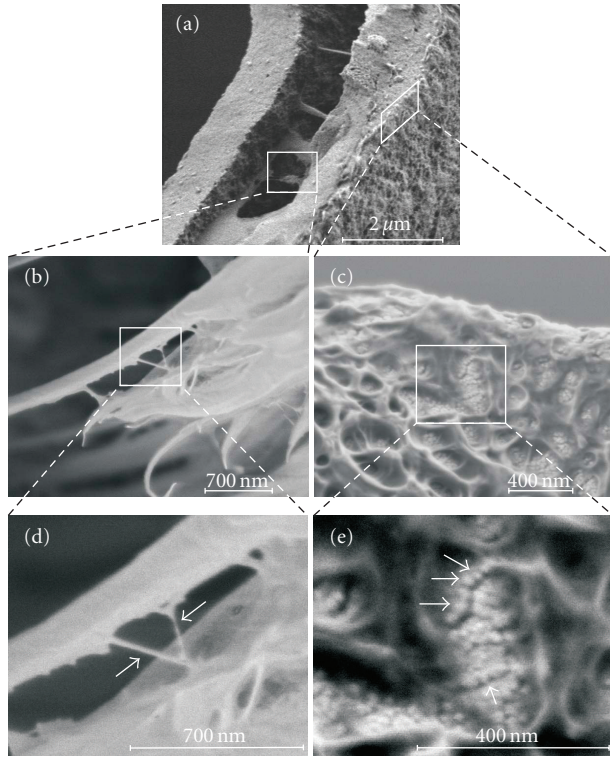


FIGURE 2: SEM images of multilayer constructed *M. chuni* spicule (a) treated with alkali solution which provides strong evidence that the multifibrillar organic matrix is the template for silica mineralization (b)–(e). Spicule layers are connected among each other by nanostructured protein fibres (arrows) (b), (d). Micrograph (e) shows a silica distribution on the surface of nanofibrils in the form of nanopearl necklets (arrows).

for further analysis using LTQ and MALDI peptide finger printings. A comparison to the MSDB protein database [27] led to the identification of collagen alpha 1 in two high MW bands. In contrast to *H. sieboldi* [9], collagen isolated and identified by the same way from *Monorhaphis sp.* was matched only to type I collagen pre-pro-alpha (I) chain (COL1A1) from dog (AAD34619) (MW 139,74). To our best knowledge, this work is the first study which confirms the presence of collagen within the spicules of *Monorhaphis* sponge and not only on their surface in the form of a collagen net which covers spicules as recently described by Müller et al. [5].

We also used highly sensitive FTIR methods for the identification of collagen isolated from spicules of *M. chuni*. Spectra obtained from this collagen, calf skin collagen type I and *C. reniformis* collagen standards were compared to each other in order to elucidate changes in protein secondary structure. The results obtained from the FTIR study (data not shown) show that collagen derived from this glass sponge exhibited spectra very similar to those from calf skin and *C. reniformis* collagens [28]. The presence of collagen fibrils in alkali solution is no surprise. Hattori et al. [29] investigated the resistance of collagen to alkali treatment at a concentration range of between 3 and 4% NaOH at 37°C in vitro. The results ob-

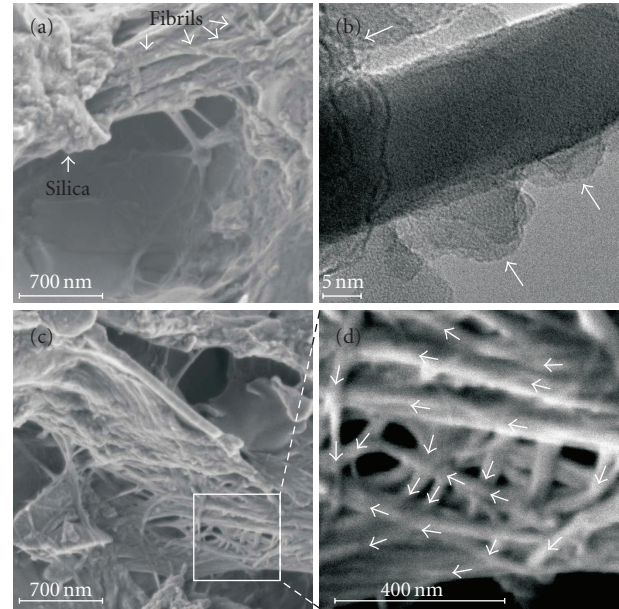


FIGURE 3: SEM and TEM nanoimagery of the fibrillar organic matrix within partially demineralized spicule. (a) Axial filament is an organization of microfibrils with a diameter of approximately 25–30 nm covered with a silica-containing layer and distributed along the axis of spicule. (b) Nanolocalization of amorphous silica particles (arrows) on the surface of partially demineralized protein fibrils using HRTEM. (c), (d) Collagen fibrils' orientation within spicule possesses a twisted plywood architecture (arrows).

tained indicated that the triple helical conformation and the helicity of the collagen molecule were maintained throughout the period of the alkaline treatment.

The procedure of alkali slow etching opens the possibility to observe the forms of collagen fibrils located within silica layers of spicules and their distribution. The results obtained by SEM observations of the desilicified spicular layers provide strong evidence that collagen fibrils' orientation within *M. chuni* spicules possesses twisted plywood architecture (Figures 3(c) and 3(d)). The twisted plywood or helicoidal structure of collagen fibrils is well-described by Giraud-Guille [30] for both in vivo and in vitro [31] systems. Spiral twisting of the collagen fibril orientation was found in several biological tissues and described for different organisms including cuticular collagens of polychaete, vestimentifera, scale collagens of primitive and bony fishes, and finally collagen fibers inside bone (all reviewed in [21]).

According to the model proposed by Giraud-Guille, adjacent lamellae have different orientations; either longitudinal (with the collagen fibers along the long axis of the lamellar sheet) or transverse (with the collagen fibers perpendicular to the long axis). From a mechanical point of view, helicoidal structures have certain advantages in resisting mechanical loads compared to orthogonal plywood structures since the twisted orientation enables a higher extensibility in tension and compression [32]. The twisted plywood architecture of collagen fibrils within basal spicules of *Monorhaphis* visible after alkali treatment (Figure 3) is very

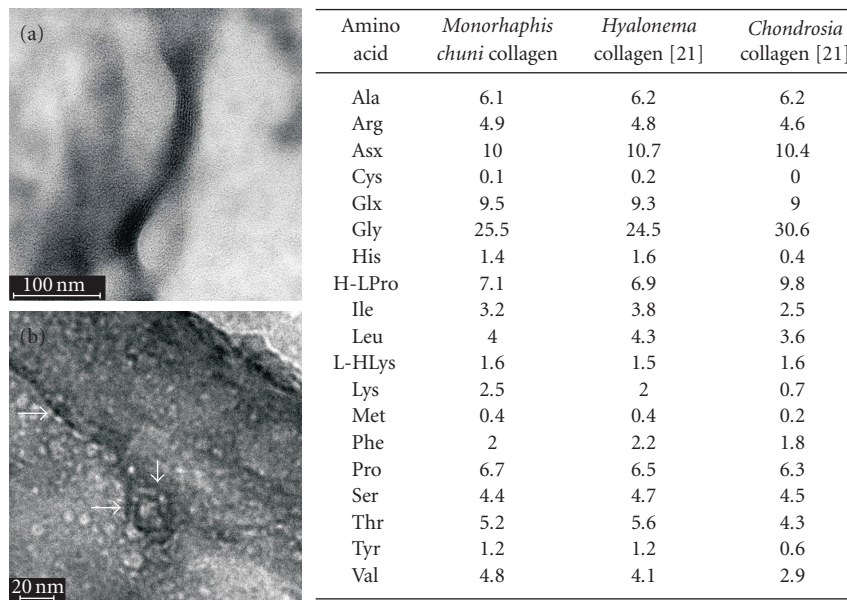


FIGURE 4: (a) High-resolution transmission electron microscopy image of the fragment of *M. chuni* collagen microfibril; (b) the arrows indicate the presence of nanofibrillar structures with a diameter which corresponds to that of collagen triple helices (1.5 nm). The results of aminoacid analysis (right) of these microfibrills showed an aminoacid content typical for collagens isolated from different sources [21].

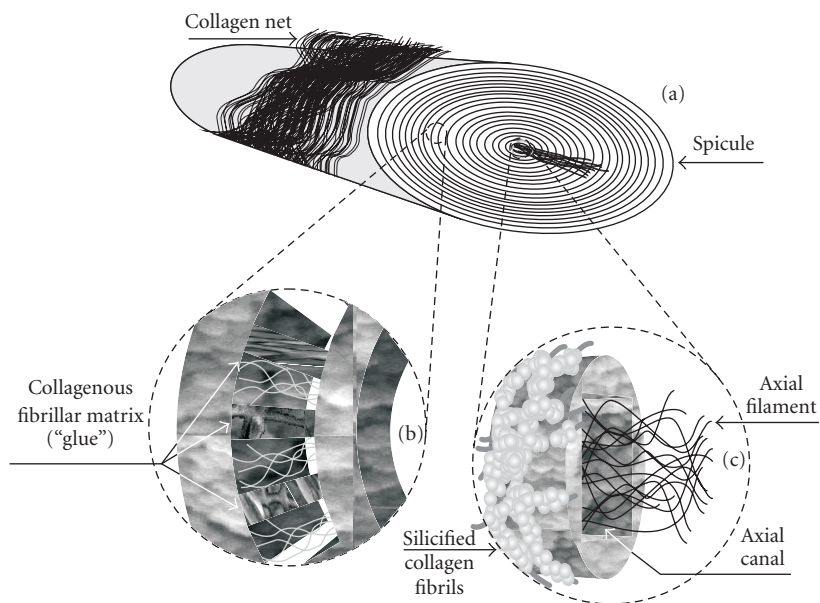


FIGURE 5: proposed model of micro- and nanostructural organisation of the basal spicule of *M. chuni* with respect to the organic matrix. (a) Collagen nets, surrounding the spicules, showed a tight mat of nanofibrils. Schematic view (b) shows a collagenous fibrillar matrix which could function as a glue between concentric layers. Image (c) represents the region of the axial canal and axial filament. The axial canal of *M. chuni* possesses a characteristic quadratic opening (c) and contains oriented bundles of unsilicified collagenous nanofibrils. The base material of the walls of the axial canal and concentric layers distributed above it consists of silicified collagen fibrils with a twisted plywood orientation. This kind of fibrillar architecture could be responsible for the remarkable micromechanical properties of the spicule as a biocomposite.

similar to that reported for lamellar bone and thus could also confirm the Girault-Guille model in the case of biosilicification in vivo. Correspondingly, this kind of collagen fibril orientation could explain why sponge spicules exhibit specific flexibility and can be bent even to a circle as reported previously [2, 4, 21]. From this point of view, basal

spicules of *Monorhaphis* sponges could be also defined as natural plywood-like silica-ceramics organized similarly to the crossed-lamellar layers of seashells [33]. Thus, we suggest that the matrix of the *M. chuni* anchoring spicule is silicified fibrillar collagen rather than collagen-containing silica which is the reason for their remarkable mechanical flexibility.

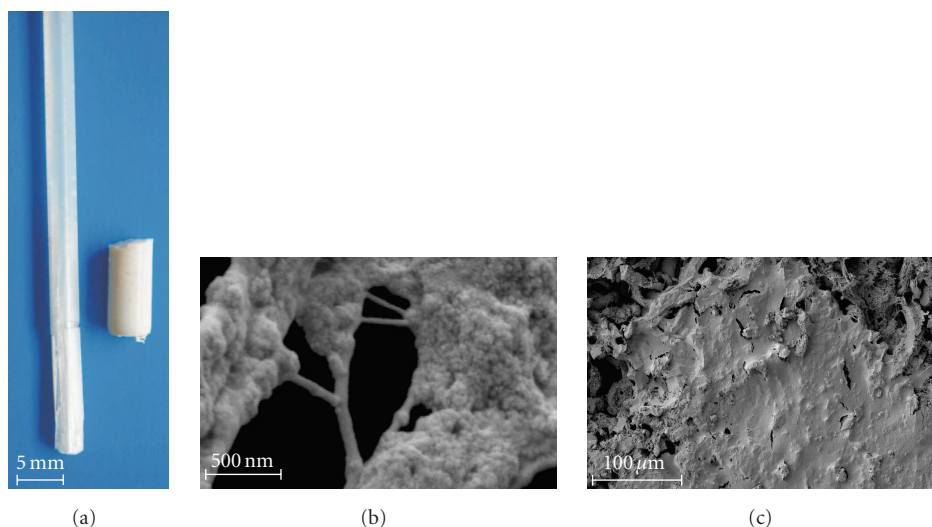


FIGURE 6: Rod-like collagen-silica-based biomaterial derived *in vitro* (a) shows morphological similarity to *M. chuni* basal spicule (a, left). SEM image (b): nanoparticles of amorphous silica deposited *in vitro* from silicic acid solution on sponge collagen fibrils replicate the nanostructure of glass sponge spicules (Figure 2(e)). SEM micrograph (c) of the surface of silica-collagen hybrid material after 14 days of cultivation of human mesenchymal stem cells, which shows high biocompatibility on this substrate.

Contrary to the postulate that silicateins, as the major biosilica-forming enzymes present in demosponges [34], are responsible for the formation of silica-based structures in all sponges, we suggested that silicateins are associated with collagen [21]. From our point of view, silicateins resemble cathepsins, which are known to be collagenolytic and capable of attacking the triple helix of fibrillar collagens. Therefore, it is not unreasonable to hypothesize that silicateins are proteins responsible for the reconstruction of collagen to form templates necessary for the subsequent silica formation. According to a dynamic model proposed by Müller and his team [5], collagen guides the silicatein(-related) protein/lectin associates concentrically along the spicules of *M. chuni*. On the basis of the results presented in this paper, we propose a model for the structure of the spicules of *Monorhaphis* sponges, including micro- and nanoaspects, which can be seen in Figure 5.

Recently, we confirmed that silicification of sponge collagen *in vitro* occurs via self-assembling, nonenzymatic mechanisms [15, 21]. To verify whether the collagenous matrix shapes the morphology of the spicules, we carried out *in vitro* experiments in which we exposed collagen to silicic acid solution ($\text{Si}(\text{OH})_4$). We obtained rod-like structures of several mm in diameter and demonstrated their similarity to the sponge spicules (Figure 6(a)). The ultrastructural analysis of these self-assembled, collagen-silica composites demonstrates that amorphous silica is deposited on the surface of collagen fibrils in the form of nanoparticle necklets (Figure 6(b)), closely resembling the nanoparticulate structure of natural *M. chuni* spicules (Figures 2(e) and 3(a)).

Bridging the nano- and microlevel, we used different techniques to create a wide spectrum of macroscopic silica-collagen-based hybrid materials. These are highly biocompatible, as demonstrated by the successful cultivation and os-

teogenic differentiation of human mesenchymal stem cells on our materials (Figure 6(c)), and potentially useful for technical and biomedical applications. On the basis of the results reported above, we also developed an advanced procedure for the biomimetically inspired production of monolithic silica-collagen hybrid xerogels [16]. The disc-like samples showed convincing homogeneity and mechanical stability, enabling cell culture experiments for the first time on such materials.

4. CONCLUSION

Recently, interest in biomaterial properties of silica-containing structures made by living sponges has grown. In order to exploit the mechanisms for the synthesis of advanced materials and devices, an investigation of the nanoscopic structure of the three-dimensional networks of these remarkable biomaterials needs to be performed [35–38]. Understanding the composition, hierarchical structure, and resulting properties of glass sponge spicules gives impetus for the development of equivalents designed *in vitro*. We showed for the first time that the silica skeletons of hexactinellids represent examples of biological materials in which a collagenous or chitinous organic matrix serves as a scaffold for the deposition of a reinforcing mineral phase in the form of silica. These findings allow us to discard different speculations about materials, which have previously been defined as organic structures (layers, filaments, surfaces) of unknown nature, and open the way for detailed studies on sponge skeletons and spicules as collagen- and/or chitin-based nanostructured biocomposites with high potential for practical applications.


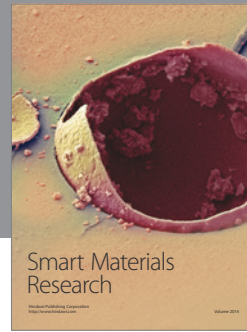
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